# Synthesis and Characterization of PPY-PANI-ZnO Hybrid Nanocomposites for OLEDs Applications as Electron Transport Layer

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# Abstract

PPY, PANI and PPY-PANI-ZnO ternary nanocomposites were synthesized in the laboratory via chemical oxidative polymerization method. The formation of PPY, PANI and corresponding change in the nanostructure with the incorporation of ZnO was examined by X-ray diffraction technique (XRD). Nano-flakes like structure were formed in the PPY-PANI polymeric matrix with the embodiment of ZnO nanoparticles, as examined by surface topology images. Optical transmittance spectra showed the optimized reduced band gap ~2.67eV. Increased rate of electron-hole nonradiative recombination associated to blue and green bands was confirmed from PL spectra. An enhanced current density ~197.75% was examined due to the significantly improved conductivity of PPY-PANI-ZnO ternary nanocomposites. This improved conductivity, reduced band gap and high electron-hole recombination rate confirmed that it can be used as electron transport layer in the OLED devices. Copyright © VBRI Press.

Keywords: PPY-PANI-ZnO nanocomposites, chemical oxidative polymerization, PL, I-V, electron transport layer.

## Introduction

Exceptionally unique characteristic of organic materials leads to the application in electronic and opto-electronic devices [1]. Polymer based organic light emitting diode is one of the most prominent application as thin film light source. It has attracted much more attention due to its many benefits such as large area luminescent display, low cost manufacturing, flexible transparent display etc. [2, 3]. In order to enhance the lifetime, efficiency and brightness, OLED's are customized as multilayer structure appending electron transport layer, emission layer and hole transport layer [4, 5]. The reflectance and transmittance of electron and hole are due to electrodes (cathode and anode). Inserted electrons and holes recombine in the emissive layer and produce excitons [6].

The embodiment of the inorganic particles into polymeric pattern can provide high performance novel materials which have enhanced structural, optical and electrical properties. Many researchers have evolved various conductive polymers with polyaromatic including polyindole, backbone polypyrrole, polythiophene, polyaniline etc. [7]. Among these various polymers polypyrrole and polyaniline have received huge interest due to their optical and electrical properties used in various opto-electronic applications. Both the polymers have high conductivity, environment stable, easy to synthesize and high electron mobility [8]. Whereas, ZnO nanoparticle have received much more attention in last two-three decades. ZnO is direct wide band gap semiconductor (3.37 eV) and has exceptional excitation binding energy ~ 60 MeV. This wide band gap is favorable for the use of ZnO to gather high-energy photons (UV light). ZnO has 90% transparency in visible region. It has also high electron transfer capability. These attractive physical and optical properties make it potential oxide for optoelectronic devices [9, 10]. In above context, ZnO is the successful candidate for the incorporation in PPY-PANI polymeric pattern. PPY-PANI based ZnO has high electron mobility as well as high electron hole recombination rate.

Many researchers have developed the investigation on the polymeric based ZnO nanocomposites. Samzadeh-Kermani et al. reported the synthesis and characterization of PVA-PANI-ZnO ternary nanocomposites prepared via In-situ polymerization method for bactericidal properties [11]. Khan et al. studied the synthesis and characterization of the PANI-ZnO nanocomposites. They also reported the stability in term of DC electrical conductivity for ammonia vapor detection application [12]. Chougule *et al.* reported the effect of CSA (camphor sulfonic acid) doping on PPY-ZnO hybrid nanocomposite for the opto-electronic application [13]. In this present work, we report the concentration dependent optical, structural and electrical performance PPY-PANI-ZnO ternary nanocomposites for the application in OLED devices as electron transport materials synthesized via chemical oxidative polymerization method.

# Experimental

### Materials

Pyrrole monomer (molecular weight = 67.09 g/mol) (Merck) with 99.99% pristine, aniline monomer (molecular weight = 93.13) (Merck), Ammonium persulfate (molecular weight = 228.18 g/mol) (Merck) and as synthesized ZnO nanoparticle were procured for the synthesis process. Deionised water was used as solvent during chemical oxidative polymerization process.

## Synthesis of polyaniline (PANI)

2 ml aniline monomer was added dropwise in 1.5 mol of HCL in 75 ml distilled water and the solution was stirred constantly. Afterwards, 4g ammonium persulfate as an oxidant was dissolved in 25 ml double distilled water and then added dropwise into the above solution. For the polymerization, the solution was maintained at temperature range of 0 - 5 °C with vigorous starring for 4 hours. The obtained dark green precipitate was collected and washed with 3 times distilled water and HCL respectively to remove the excess oxidant and monomer. Lastly collected precipitate was dried in vacuum oven at 70 °C for 24 h.

## Synthesis of PPY-PANI-ZnO ternary nanocomposites

PPY-PANI-ZnO ternarv nanocomposites were successfully prepared via chemical oxidative polymerization process. Firstly, 0.1M of pyrrole monomer was dissolved in 75ml of distilled water. Afterwards ZnO was added in varying concentration (5%, 10% and 15%) and exact half amount of PANI in comparison to pyrrole monomer was also added in above solution. Moreover, 0.1M of ammonium persulfate (APS) was dissolve in the 25ml of distilled water and added dropwise in the above monomer solution. For the polymerization whole solution setup was maintained at temperature range of 0 - 5°C under vigorous starring for 4-5h. The obtained dark blackgreen precipitate was washed with distilled water and ethanol respectively. Finally, the collected precipitate was dried in vacuum chamber at 70 °C for 24 hours. The schematic representation for the synthesis of PPY-PANI-ZnO ternary nanocomposites has been shown in Fig. 1.



Fig.1. Schematic diagram for the synthesis of PPY-PANI-ZnO ternary nanocomposites.

### Characterization

The as synthesized PPY, PANI and PPY-PANI-ZnO ternary nanocomposites were characterized by X-ray diffractometer (XRD-GBCMMA, Cu-Ka radiation,  $\lambda = 1.5406$  Å) in the 2 $\theta$  angular range  $10^{\circ} - 65^{\circ}$ . The surface topology and EDX analysis of the ternary nanocomposites were studied by Field emission scanning electron microscopy (ZEISS, SUPRA-55) and Energy Dispersive Microanalysis (Oxford Liquid Nitrogen free SDD X MAX 50 EDS) respectively. Optical transmittance spectrum obtained by UV-Vis spectroscopic technique (CARY-5000) in the wavelength range of 200nm - 800nm. Photoluminescence (PL) spectroscopy was performed by fluorescence spectrophotometer (Hitachi F-2500 Fluorescence Spectrophotometer) at the excitation wavelength of 320 nm in the emission range of 300 nm-600 nm. The current-voltage measurements of these ternary nanocomposites were recorded using source meter (KEITHLEY 2400).

## **Results and discussion**

## XRD analysis

XRD spectra of Pure PPY, PANI and PPY-PANI-ZnO nanocomposites have been shown in the **Fig. 2a** – **Fig. 2e**. Broad diffraction peak located at 23.15 for PPY in **Fig. 2a** indicating amorphous nature. In **Fig. 2b**, diffraction spectrum of PANI showed the peaks at  $2\theta = 23.66^{\circ}$  assigned to the periodicity perpendicular to the polymer chains, revealing the degree of amorphous nature [14].



Fig. 2. XRD pattern for (a) PPY, (b) PANI, (c-e) PPY-PANI-ZnO nanocomposites of varying concentration.

**Fig. 2c - Fig. 2e** indicates the diffraction pattern of PPY-PANI-ZnO nanocomposites comprised a broad peak for polymers at  $21^{\circ}$  and sharp peaks at an angle  $2\theta = 31.99^{\circ}$ ,  $34.94^{\circ}$ ,  $36.76^{\circ}$ ,  $47.86^{\circ}$ ,  $57.10^{\circ}$  and  $63.16^{\circ}$  corresponding to  $(1\ 0\ 0)$ ,  $(0\ 0\ 2)$ ,  $(1\ 0\ 1)$ ,  $(1\ 0\ 2)$ ,  $(1\ 1\ 0)$  and  $(1\ 0\ 3)$  for ZnO nanoparticle, well matched with

JCPDS database (Card no. 36-1451) confirmed the semi-crystalline nature. The average crystallite size of nanocomposites was calculated by Debye Scherrer's formula ( $D = \frac{K\lambda}{BCos\theta}$ ) to be 45 nm and it decreased with increase in ZnO content. The average inter chain separation ( $S = \frac{5\lambda}{8Sin\theta}$ ) was calculated ~ 7.25 Å. The diffraction peaks of corresponding nanocomposites confirmed the formation of PPY-PANI-ZnO ternary nanocomposites.

#### **FESEM** analysis

In order to investigate the surface topology and morphology, FESEM analysis has been carried out. Fig. 3 showed the FESEM images of pristine PPY, PANI and PPY-PANI-ZnO ternary nanocomposites of varying concentration. Pristine PPY showed the spherical ball type structure while pristine PANI exhibited agglomerated nanofibers. FESEM images of ternary nanocomposite showed, with the addition of ZnO content in ternary nanocomposite, nano-flakes structure formed and it also increased the aggregation of the nanoparticle due to the absorption of H atom from NH<sub>2</sub> group of polypyrrole on the surface of ZnO nanoparticle. This confirmed the increment of crystallinity with the addition of the ZnO content. The EDX (energy dispersive X-ray spectroscopy) spectra of PPY-PANI-ZnO ternary nanocomposite confirmed the successful embodiment of ZnO nanoparticle into polymeric pattern.



**Fig. 3.** FESEM images for (a) PPY, (b) PANI, (c) 5% PPY-PANI-ZnO, (d) 10% PPY-PANI-ZnO, and (e) 15% PPY-PANI-ZnO nanocomposites and EDX spectrum of nanocomposite sample.

#### UV-Vis transmittance spectroscopy

The transmittance spectra of pristine PPY, PANI and PPY-PANI-ZnO ternary nanocomposites were reported

using double beam UV-Visible spectrometer and shown in Fig. 4. These ternary films had excellent transmittance and very low absorption and reflectance. It showed 80% transparency in the visible region. There was almost negligible significant change in the transmittance spectrum observed for pure PPY, PANI and their nanocomposites in the wavelength range 200 nm - 250 nm due to the high ultraviolet light absorption (low transmittance). This low transmittance corresponded to the PPY and PANI organic fillers. The transmittance spectrum of ternary nanocomposites was increased from the 350 nm, probably due to the reduced absorption of ZnO nanoparticle. The optimized higher band edge transmittance was observed for the 10% PPY-PANI-ZnO nanocomposites around 490 nm associated to blue-green region.



Fig. 4. UV-Visible transmittance spectra for PPY, PANI and PPY-PANI-ZnO nanocomposites of varying concentration.

#### Band gap measurement

The optical band gap of the PPY, PANI and PPY-PANI-ZnO ternary nanocomposites was determined from the transmittance spectrum shown in **Fig. 5**. The frequency dependent absorption coefficient stated in following relation;

$$(ahv)^{1/n} = A(hv - E_g) \tag{1}$$

where,  $\alpha$ -absorption coefficient, hv- incident photon energy, A- constant, E<sub>g</sub> - band gap and n- nature of electronic transition. The absorption coefficient  $\alpha$  is calculated using Beer's Lambert law stated in following relation.

$$\alpha = \frac{\ln(\frac{1}{T})}{d} \tag{2}$$

where, T represents transmittance of the films and d represents the film thickness. The as prepared PPY, PANI and PPY-PANI-ZnO ternary nanocomposites were deposited in the glass substrate by spin coating technique. The direct allowed plot between  $(\alpha hv)^2$  and hv of ternary nanocomposites revealed the huge influence due to incorporation of ZnO nanoparticle. It was observed that with the increase in ZnO concentration in PPY host matrix, optical energy band gap decreased due to the formation of new level in the

energy gap, which begun to move the electron from valance band through new level to conduction band. Subsequently, band gap decreased and conductivity increased. The optimized band gap of 10% PPY-PANI-ZnO ternary nanocomposites was  $\sim 2.67$  eV. The reduction in band gap led to increased p-type conductivity.



**Fig. 5.** Optical band gap for (a) PPY, (b) PANI, (c) 5% PPY-PANI-ZnO, (d) 10% PPY-PANI-ZnO, and (e) 15% PPY-PANI-ZnO nanocomposites.

#### Photoluminescence analysis

The PL spectra emission spectra for PPY-PANI-ZnO ternary composites at 320 nm excitation energy have been shown in Fig. 6. There were two distinguishable emission peaks observed at 450 nm and 495 nm due to nonradiative recombination of electron-hole pair in the nanocomposite film. These exhibited the luminescence under blue and green region respectively. The additional peak recorded at 465 nm was attributed to deep energy states caused by impurities of ZnO filler in the ternary nanocomposite. The emission peak observed at 450 nm due to the phonon assisted transition accredit due to recombination between the electrons in deep defect level and holes in valance band of PPY-PANI-ZnO ternary nanocomposites [15, 16]. The green band occurred due non-radiative recombination of electron in the conduction band and holes trapped by the site of oxygen (O2) vacancy of ZnO fillers. PL spectra showed the optimum luminance intensity for the 10% PPY-PANI-ZnO ternary nanocomposites due to the high electron-hole recombination rate. The increase rate of electron-hole recombination is well in favor of OLED's as electron transporting layer.



Fig. 6. PL spectra for PPY-PANI-ZnO nanocomposites of varying concentration.

#### J-V measurement

Fig. 7 showed the J-V Characteristics of Pristine PPY, PANI and PPY-PANI-ZnO ternary nanocomposites. The measurement was taken at room temperature in the potential range of -8.0 V to +8.0 V. All recorded data clearly indicates the linear relation between current and voltage which confirmed the ohmic nature. Current density was increased with the increase ZnO content and optimized current density was found for 10% PPY-PANI-ZnO ternary nanocomposites ~ 7.95 mA/cm<sup>2</sup> due to the higher electron mobility in nanocomposites. An enhanced current density was recorded at 197.75 % as compared to pure PANI. Based on above clarification electron-hole recombination was carried out. Embodiment of ZnO nanoparticle into polymeric pattern increased the minority charge carrier (hole) as compared to the majority charge carrier (electron). Consequently resistivity decreased and recombination PPY-PANI-ZnO was higher in 10% rate nanocomposites showing the p-type behavior which was in good agreement to the result of UV-Vis absorption spectroscopy. These electrical properties confirmed that this ternary nanocomposite can be used as electron transport layer in OLED devices.



Fig. 7. J-V characteristics for PPY, PANI and PPY-PANI-ZnO nanocomposites of varying concentration.

#### Conclusion

Pristine PANI, PPY and PPY-PANI-ZnO ternary nanocomposites of varying concentration were successfully synthesized via chemical oxidative polymerization method. The average crystallite size and average chain separation was ~ 45 nm and ~ 7.25 Å respectively and corresponding diffracted peaks confirmed the formation of PPY-PANI-ZnO ternary nanocomposites. FESEM images showed the nanoflake structure formed due to ZnO nanofillers in the surface PPY-PANI. EDX analysis confirmed the successful incorporation of ZnO in polymeric pattern. The UV-Vis transmittance spectroscopy showed ~ 80% transparency in visible region and transmittance characteristic peak at 490 nm corresponded to bluegreen region. Optimum band gap ~ 2.67 eV for 10% PPY-PANI-ZnO nanocomposites. PL emission spectra showed the two identifiable peaks at 450 nm and 495 nm associated to the blue band and green band respectively. An enhance current density ~ 197.75 % as compared to PANI neat polymer was found for the 10% PPY-PANI-ZnO ternary nanocomposites. These optical and electrical properties confirmed its application in OLED device fabrication as electron transport materials.

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